Provided for non-commercial research and education use. Not for reproduction, distribution or commercial use.



This article appeared in a journal published by Elsevier. The attached copy is furnished to the author for internal non-commercial research and education use, including for instruction at the authors institution and sharing with colleagues.

Other uses, including reproduction and distribution, or selling or licensing copies, or posting to personal, institutional or third party websites are prohibited.

In most cases authors are permitted to post their version of the article (e.g. in Word or Tex form) to their personal website or institutional repository. Authors requiring further information regarding Elsevier's archiving and manuscript policies are encouraged to visit:

http://www.elsevier.com/copyright

Solid State Ionics 192 (2011) 21-24

Contents lists available at ScienceDirect



Solid State Ionics

journal homepage: www.elsevier.com/locate/ssi

## Water as a source for oxygen incorporation into acceptor doped SrTiO<sub>3</sub> single crystals

Christos Argirusis <sup>a,b,\*</sup>, Jana Grosse-Brauckmann <sup>b</sup>, Pradyot Datta <sup>b</sup>, Francois Jomard <sup>c</sup>, Florian Voigts <sup>d</sup>, Wolfgang Maus-Friedrichs <sup>d,e</sup>

<sup>a</sup> School of Chemical Engineering, National Technical University of Athens, Iroon Polytechneiou 9, Zografou 157 80, Athens, Greece

<sup>b</sup> Institut für Metallurgie, Technische Universität Clausthal, Robert-Koch-Straße 32, 38678 Clausthal-Zellerfeld, Germany

<sup>c</sup> GEMaC, UMR 8635/CNRS Université de Versailles-Saint-Quentin, 1 place A. Briand, 92195 Meudon Cedex, France

<sup>d</sup> Institut für Physik und Physikalische Technologien, Technische Universität Clausthal, Leibnizstrasse 4, 38678 Clausthal-Zellerfeld, Germany

<sup>e</sup> Clausthaler Zentrum für Materialtechnik, Walther-Nernst-Straße 7, 38678 Clausthal-Zellerfeld, Germany

#### ARTICLE INFO

Article history: Received 31 August 2009 Received in revised form 8 April 2010 Accepted 3 May 2010 Available online 2 June 2010

Kevwords: Strontium titanate Oxygen incorporation Surface exchange coefficient Perovskite Water dissociation

## 1. Introduction

SrTiO<sub>3</sub> is a model oxide for electroceramics. It is a typical ABO<sub>3</sub> type perovskite material with remarkable thermal and chemical stability and easy fabricability. As a result of that SrTiO<sub>3</sub> has found applications in a wide range of technologically relevant fields like capacitors, oxygen sensors, dielectrics as well as substrate for superconducting materials. Moreover, depending on the conditions it can show ionic conduction or p- or n-type electronic conduction. The key point that enables one to render this material with a specific conductivity is its suitability for doping with aliovalent dopants without changing its crystal structure [1,2].

The defect chemical basics of this material can be found in the literature [3,4] and in the references therein. For many applications, the diffusion of oxygen vacancies is the key property. It is responsible for the degradation of multilayer capacitors at room temperature [5], but is also the basic principle for the application of SrTiO<sub>3</sub> as resistive high temperature oxygen sensor [6,7], with the analysis of combustion engine exhaust being one important utilisation. The incorporation of oxygen from the gas phase into the crystal is a process involving multiple steps, from the dissociation of oxygen containing molecules in the vicinity of the surface followed by the adsorption at the surface to the oxygen diffusion in the bulk. Recently Merkle and

## ABSTRACT

The possibility of the incorporation of oxygen and deuteron stemming from the dissociation of water into iron doped strontium titanate single crystals with two different iron concentrations (0.13 and 0.013 at.% Fe) has been investigated. Double labeled water (<sup>2</sup>H<sub>2</sub><sup>18</sup>O) was used for Isotope Exchange Depth Profiling. It acts as an incorporation source of  $^{18}$ O into Fe-doped SrTiO<sub>3</sub>(100) single crystals. Total dissociation of the water takes place in the vicinity of the SrTiO<sub>3</sub> surface and both oxygen and deuteron incorporate into the SrTiO<sub>3</sub>. The results of the diffusion profile evaluation confirm that oxygen and deuteron migrate independently in the bulk. Oxygen incorporates into the material at high incorporation rates, practically comparable with the ones resulting from an O2 atmosphere. The interaction of H2O with SrTiO3(100) surfaces was further studied with Metastable Induced Electron Spectroscopy and Ultraviolet Photoelectron Spectroscopy. Neither partial dissociation of water molecules nor any significant formation of surface OD groups was observed.

© 2010 Elsevier B.V. All rights reserved.

OLID STATE IONIC

Maier [8] discussed the oxygen incorporation into oxides on the example of SrTiO<sub>3</sub> as a model material. The rate determining step in the total reaction is the surface interaction.

The most convenient way of studying oxygen incorporation in a crystalline solid is by means of tracer isotope exchange. In this process the oxygen incorporation consists of a surface reaction at the gas solid interface which is characterized by the surface exchange rate constant and a subsequent bulk transport which is characterized by the diffusion coefficient. The objective of this investigation is to see whether different gaseous contaminants and in particular water has an influence on the oxygen incorporation reaction in acceptor doped SrTiO<sub>3</sub>. Results for oxygen incorporation from <sup>2</sup>H<sub>2</sub><sup>18</sup>O sources are reported. The results are supported by measurements with surface sensitive methods like Ultraviolet Photoelectron Spectroscopy (UPS) and Metastable Induced Electron Spectroscopy (MIES).

#### 2. Experimental

Single crystals of SrTiO<sub>3</sub>(100) were supplied by CrysTec GmbH (Berlin, Germany). First, each of the two differently doped SrTiO<sub>3</sub> single crystals (0.13 and 0.013 at.% Fe) were cut from one boule each with the dimension of  $5 \times 5 \times 1$  mm<sup>3</sup> and have been polished down to optical finish with the dimension 0.1 µm.

For the diffusion experiments, all samples were heated at 1000 °C for 24 h in 200 mbar of  ${}^{16}\text{O}_2$  in order to heal possible surface defects stemming from the polishing procedure.

<sup>\*</sup> Corresponding author. School of Chemical Engineering, National Technical University of Athens, Iroon Polytechneiou 9, Zografou 157 80, Athens, Greece. E-mail address: christos.argirusis@tu-clausthal.de (C. Argirusis).

<sup>0167-2738/\$ -</sup> see front matter © 2010 Elsevier B.V. All rights reserved. doi:10.1016/j.ssi.2010.05.005

The samples were afterwards equilibrated at the respective temperature for about ten times the estimated subsequent diffusion time in argon, meaning at an oxygen partial pressure similar to the subsequent diffusion from water in order to equilibrate the defect concentration. Two sets of samples were then annealed separately in water atmospheres with different double labeled water ( $^{2}H_{2}^{18}O$  from Campro Scientific, Berlin, Germany) concentration at various temperatures between 850 and 1000 °C. Subsequently, oxygen incorporation processes were studied by Secondary Ion Mass Spectroscopy (SIMS). A detailed description of the procedure for this Isotope Exchange Depth Profiling (IEDP) can be found in [9].

The interaction of H<sub>2</sub>O with SrTiO<sub>3</sub>(100) surfaces was further studied with MIES and UPS. Electron spectroscopy is performed in an ultrahigh vacuum (UHV) apparatus with a base pressure of  $5 \times 10^{-11}$  mbar using a hemispherical analyzer (VSW HA100) in combination with a source for metastable helium atoms (He<sup>\*</sup> <sup>3</sup>S<sub>1</sub>) and ultraviolet photons (HeI line). Details on the apparatus used and the theoretical background may be found in [10].

MIES is an extremely surface sensitive technique probing solely the outermost layer of the sample, because the He<sup>\*</sup> atoms interact with the surface typically 0.3 to 0.5 nm in front of it. This may occur via a number of different mechanisms depending on surface electronic structure and work function, which are described in detail elsewhere [11–13]. On SrTiO<sub>3</sub> surfaces a special Auger Deexcitation (AD) type interaction occurs [14]: The 2s electron of the impinging He<sup>\*</sup> is resonantly transferred into the surface of the sample and localizes at near surface Ti 3d states. Subsequently, a Ti 3d electron fills the hole in He<sup>+</sup> 1s in an interatomic Auger neutralization (AN) process, followed by the emission of an O 2p surface electron carrying the excess energy.

#### 3. Results

Fig. 1 shows the results of a MIES/UPS experiment on a SrTiO<sub>3</sub> surface with low iron content (0.013 at.% Fe). Prior to the experiment, the surface was annealed in UHV at about 970 K for 120 min. to remove contaminations from the surface and to produce oxygen vacancies  $V_{0}^{\circ}$ . This is done in order to achieve a sufficient conductivity in the crystal to prevent charge-up during electron spectroscopy. Subsequently, water is offered to the surface at an exposure of 50 Langmuir (L) per spectrum, one Langmuir equals  $1.33 \times 10^{-4}$  Pa s. The

change in the MIES/UPS is shown in a waterfall manner, indicated by the arrow on the right side.

The electron count rate measured during the experiment is plotted versus the electron binding energy, 0 eV corresponds to the Fermi energy. The lowermost spectrum shows the cleaned surface. In UPS, the density of states of the valence band region of  $SrTiO_3$  is detected. Due to the pinning of the Fermi level to the conduction band, the gap energy of 3.2 eV can be recognized, as well as two distinct features at 5 eV and 7 eV. The two features correspond to emission from O 2p-derived orbitals in the surface, the one at 7 eV corresponding to emission from orbitals hybridized with Ti 3d states; the one at 5 eV arises from ionization of non-hybridized orbitals.

He<sup>\*</sup> during MIES interact with the surface as described in the experimental section. Only one peak at about 6.5 eV is detectable. It arises from the ionization of non-hybridized O 2p-derived orbitals, because these orbitals are present at the surface due to the broken translation symmetry and protrude much wider into the vacuum as every other state. Therefore He<sup>\*</sup> preferentially interact with these states. Thus, the MIES detects the density of the uppermost surface states only.

Fig. 2 gives a clear impression of the concentration of  $^{18}$ O in the bulk of the SrTiO<sub>3</sub> single crystals stemming from the dissociation of the double labeled water. The SIMS diffusion profiles presented in Fig. 3 show oxygen profiles but not any profiles for deuterons at all.

### 4. Discussion and conclusions

Both the MIES and the UPS spectra (Fig. 1) show the well known valence band of  $SrTiO_3$  as expected from literature [14].

During water exposure no significant change in the density of states of the surface is detected neither in MIES nor UPS even at high exposures up to 1250 L. At the end of the experiment only the development of a very slight shoulder around 11 eV is observed. This structure may be attributed to surface OD groups from the partial dissociation of the surrounding water. Because of the very low count rate of the structure, the coverage of the surface with OD groups must be very small. Because of this, the partial dissociation of water molecules and the subsequent adsorption at the surface must be an ineffective process. Otherwise, MIES would detect much more surface OD groups.

A more detailed investigation of the surface interaction between water and  $SrTiO_3(100)$  may be found in [15]. Besides this slight OD generation on the surface, no further adsorbates could be detected.



Fig. 1. MIES (left) and UPS (right) spectra of a  $SrTiO_3(100)$  surface with low iron content (0.013 at.% Fe) during exposure to water.

C. Argirusis et al. / Solid State Ionics 192 (2011) 21-24



Fig. 2. Depth profiles of oxygen stemming from the dissociation of water in SrTiO<sub>3</sub> single crystals with low iron content (0.013 at.%, left) and high iron content (0. 13 at.%, right).

The results of the diffusion profile evaluation (Fig. 3) confirm that <sup>2</sup>H<sub>2</sub><sup>18</sup>O acts as an incorporation source of <sup>18</sup>O into the Fe-doped crystals and that oxygen and deuteron migrate independently in the bulk. This oxygen incorporates into the material at high incorporation rates, practically comparable with the ones resulting from an oxygen atmosphere [18].

Surface hydrogen and surface oxygen are not necessarily detectable by means of MIES/UPS leading to the assumption that a total dissociation of the water in the vicinity of the SrTiO<sub>3</sub>(100) surface is the source of the <sup>18</sup>O. After dissociation both oxygen and deuteron may incorporate into the SrTiO<sub>3</sub> and migrate independently in the bulk.

The low concentration of deuterons in the depth profile in Fig. 3 is a confirmation that <sup>18</sup>O-ions and deuterons migrate in the SrTiO<sub>3</sub> bulk independently, otherwise parallel depths profiles should be found. This can indicate one or both of the two following possibilities:

- 1. The deuterons do not incorporate into SrTiO<sub>3</sub> to the same extent as oxygen does, meaning that a recombination of the deuterons to  $^2\text{D}_2\text{-molecules}$  on the SrTiO\_3 surface occurs, followed by their desorption into the gas phase.
- 2. The deuterons are incorporating and diffusing much quicker than <sup>18</sup>O-ions so that they can reach the surface again, confirming that <sup>18</sup>O-ions and deuterons migrate in the SrTiO<sub>3</sub> bulk independently.



Fig. 3. Raw data of the depth profiles in SrTiO<sub>3</sub> single crystals doped with 0.13 at.% Fe at 500 °C from the dissociation of double labeled water. It is obvious that in contrast to the <sup>18</sup>O there is no deuteron diffusion profile in the bulk. The signal is practically as low as the background.

Our results up to now confirm the second possibility and thus the findings of Kreuer [16] that deuteron migrates much quicker than oxygen in SrTiO<sub>3</sub>. That means it migrates as  $^{2}D^{+}$  and not in the form of OD groups.

Based on the MIES/UPS results as well as the SIMS diffusion profiles we assume that a total dissociation of the water takes place in the vicinity of the  $SrTiO_3(100)$  and that subsequently both oxygen and deuteron incorporate into the SrTiO<sub>3</sub> and migrate independently in the bulk. This assumption is supported by the results of the evaluation of the SIMS diffusion profiles, where we found oxygen profiles (Fig. 2) but not any profiles for deuteron at all (Fig. 3). In the case of coupled migration of deuteron together with oxygen one would expect parallel diffusion profiles for oxygen and deuteron which is not confirmed with our results. Furthermore, we do not observe any significant formation of surface OD groups from the partial dissociation of water with our MIES spectrometer. Because of this, diffusion of OD groups into the bulk seems unlikely.

We can conclude that water acts as a source for oxygen incorporation into SrTiO<sub>3</sub> and because of the model character of SrTiO<sub>3</sub> probably in other perovskite materials as well. Oxygen incorporates into the SrTiO<sub>3</sub> single crystal bulk at high incorporation rates [17], practically comparable with the ones resulting from an  $O_2$ atmosphere. The influence of the dopant content and of the temperature on the oxygen diffusion in the crystal is discussed in detail elsewhere [18]. Based on these findings, we can state that attention has to be paid when such materials are used in humid atmospheres for example as sensor materials.

#### Acknowledgements

We gratefully acknowledge the financial support by the Deutsche Forschungsgemeinschaft (DFG) under contract numbers Ar 248/3-2 and Ma 1893/9-2.

#### References

- R. Moos, T. Bischoff, W. Menesklou, K.H. Härdtl, J. Mat. Sci. 32 (1997) 4247.
- S. Steinsvik, R. Bugge, J. Gjønnes, J. Taftø, T. Norby, J. Phys. Chem. Solids 58 (1997) 969.
- G.M. Choi, H.L. Tuller, J. Am. Ceram. Soc. 71 (1988) 201. [3]
- R. Moos, K.H. Härdtl, J. Am. Ceram. Soc. 80 (1997) 2549. [4]
- R. Waser, T. Baiatu, K.H. Härdtl, Mater. Sci. Eng. A 109 (1989) 171.
- W. Menesklou, H.J. Schreiner, K.H. Härdtl, E. Ivers-Tiffée, Sensors and Actuators B [6] 59 (1999) 184.
- R. Meyer, R. Waser, Sensors and Actuators B 101 (2004) 335. [7] R. Merkle, J. Maier, Angew. Chem. Int. Ed. 47 (2008) 2. [8]
- Chr. Argirusis, S. Wagner, W. Menesklou, C. Warnke, T. Damjanovic, G. Borchardt, [9]
- Ellen Ivers-Tiffée, Phys. Chem. Chem. Phys. 7 (2005) 3523 [10] F. Bebensee, F. Voigts, W. Maus-Friedrichs, Surf. Sci. 602 (2008) 1622.

# Author's personal copy

C. Argirusis et al. / Solid State Ionics 192 (2011) 21-24

- [11] Y. Harada, S. Masuda, H. Ozaki, Chem. Rev. 97 (1997) 1897.
  [12] H. Morgner, Adv. Atom. Mol. Opt. Phys. 42 (2000) 387.
  [13] G. Ertl, J. Küppers, Weinheim, VCH Verlag, 1985.
  [14] W. Maus-Friedrichs, M. Frerichs, A. Gunhold, S. Krischok, V. Kempter, G. Bihlmayer, Surf. Sci. 515 (2002) 499.
  [15] F. Voigts, Chr. Argirusis, W. Maus-Friedrichs, Surface and Interface Analysis, submitted.

- [16] K.-D. Kreuer, Solid State Ionics 125 (1999) 285.[17] Chr. Argirusis, J. Grosse-Brauckmann, F. Voigts, W. Maus-Friedrichs, in preparation.
- [18] Chr. Argirusis, F. Voigts, P. Datta, J. Grosse-Brauckmann, W. Maus-Friedrichs, Phys. Chem. Chem. Phys. 11 (2009) 3152.